# Low temperature rf-sputtered thermochromic VO<sub>2</sub> films on flexible glass substrates

Emmanouil Gagaoudakis<sup>1, 2\*</sup>, Giannis Michail<sup>2</sup>, Elias Aperathitis<sup>2</sup>, Ioannis Kortidis<sup>2</sup>, Vassilios Binas<sup>2</sup>, Marianthi Panagopoulou<sup>3</sup>, Yannis S. Raptis<sup>3</sup>, Dimitris Tsoukalas<sup>3</sup>, George Kiriakidis<sup>1,2</sup>

<sup>1</sup>Physics Department, University of Crete, P.O. Box 2208, 71003 Heraklion, Crete, Greece
 <sup>2</sup>Institute of Electronic Structure & Laser (IESL), Foundation for Research and Technology (FORTH) Hellas, P.O. Box 1385, Heraklion 70013, Crete, Greece
 <sup>3</sup>School of Applied Mathematical and Physical Sciences, National Technical University of Athens, GR 157 80, Zografou Campus, Athens, Greece

\*Corresponding author: Tel: (0030) 2810-391272; E-mail: mgagas@iesl.forth.gr

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# Abstract

The high deposition temperature of the order of 400 °C and more is requirement for the growth of the thermochromic phase of vanadium dioxide (VO<sub>2</sub>), limits the type of substrates that one may grow them on only to rigid ones. In this work, thermochromic VO<sub>2</sub> films were successfully deposited on flexible Corning® Willow® glass substrates, without the use of a buffer layer, by rf sputtering at a substrate temperature of 300 °C, one of the lowest for this technique ever reported. The critical transition temperature of 80 nm thin films was  $T_c = 50.7$  °C, transmittance hysteresis width was  $\Delta T = 12.1$  °C, while the modulation of the transmittance at  $\lambda = 2000$  nm measured at 25 °C and 90 °C was around 36%, leading to a solar modulation of  $\Delta Tr_{sol} = 5\%$ . In addition, an increase in transmittance at  $\lambda = 600$ nm (visible region) of 4% was observed before and after heating, while integrated luminous transmittance remained almost constant at  $Tr_{lum} = 34\%$ . The thermochromic and luminous characteristics of the VO<sub>2</sub> films deposited on flexible glass substrates by sputtering technique opens up a new window for thermochromic applications on flexible substrates. Copyright © 2017 VBRI Press.

Keywords: Thermochromic VO<sub>2</sub>, rf sputtering, flexible glass substrate, low deposition temperature.

# Introduction

Chromogenics are materials which change their optical properties due to an external stimulation, such as light (photochromic materials), voltage (electrochromic materials) or temperature (thermochromic materials) [1]. Among them, thermochromic materials are of great interest since they can be used as coatings for smart windows in energy saving buildings, in order to regulate the internal temperature [2, 3]. VO<sub>2</sub> is the most well studied thermochromic material because its critical transition temperature (T<sub>c</sub>) is the closest to room temperature [4-6]. In specific, VO2 undergoes a first-order semiconductor to metal transition (SMT) at a critical transition temperature (T<sub>c</sub>) of 68 °C, which can be attributed either to strong electron correlations (Mott-Hubbard transition) or to electron-lattice correlations (Peierls transition) [5, 7-8]. Below T<sub>c</sub> it is an insulator, having a monoclinic structure and high transmittance in infrared irradiation, while above T<sub>c</sub> it turns to metal, with a rutile structure and high reflectance in infrared [9]. In addition, during this procedure the visible transmittance remains constant [10].

Various deposition techniques such as APCVD [11], sol-gel [12], PLD [13], dc or rf sputtering [9, 14-17] etc., are used to growth thermochromic VO<sub>2</sub>. Among them rf sputtering is a widely-used technique in order to growth films of high quality and homogeneity, even in large scale. Thermochromic VO<sub>2</sub> films can also be deposited on a wide range of rigid substrates such as glass [13, 18], Si [18], sapphire [19], while in some cases a buffer layer of SnO<sub>2</sub> [20, 21], ZnO [21], or other metal oxides [22] may be used to obtain VO<sub>2</sub> with improved thermochromic properties. However, depositing thermochromic VO<sub>2</sub> on flexible substrates is of great interest and remains a challenge for application in retrofitting in buildings as well as other potential applications that a flexible substrate may be used like consumers electronics, automobiles etc. The relative high substrate temperature (over 300 °C) which is normally required to produce the  $VO_2$  phase by sputtering technique [14-16], works as a barrier for depositions on flexible substrates such as polyethylene terephthalate (PET) and polyethylenenaphthalate (PEN). This could be probably achieved only by methods applying chemical solution processes [23, 24]

by which, initial VO<sub>2</sub> produced in powder form and fired at high temperatures undergoes a specific treatment that turns it into a film, however with less uniformity comparable to that produced by sputtering. In this work, thermochromic VO<sub>2</sub> films were deposited on flexible Corning® Willow® glass substrates for first time, without the need of buffer layer, using the rf sputtering technique, at a substrate temperature of 300 °C which is one of the lowest reported in the literature [**17**, **25**]. The thermochromic properties of the VO<sub>2</sub> films were similar to those produced on rigid glass substrates with or without a buffer layer.

## **Experimental**

VO<sub>2</sub> films were growth by rf sputtering technique, using a vanadium metal target (8" dia. x 0.250", purity 99.95%). The total pressure during deposition was kept at 5 mTorr, while sputtering power was 400 W. The O<sub>2</sub> content in Ar / O<sub>2</sub> plasma was 1% and the substrate temperature was 300 °C, selected as the optimum conditions to produce thermochromic VO<sub>2</sub>. The films were deposited on flexible Corning® Willow® glass, using the rf sputtering deposition technique. The thickness of flexible glass was 0.2 mm, while that of VO<sub>2</sub> films was around 80 nm as verified by both a-step profilometer and Field Emission Scanning Electron Microscopy (FE-SEM) measurements. Film structure was examined by X-Ray Diffraction (XRD) technique using a Panalytical X' Pert Diffractometer system with Cu  $K_{\alpha}$  X-Rays. Grazing Incidence XRD (GIXRD) method with  $\theta = 0.5^{\circ}$  and  $2\theta =$  $10^{\circ} - 80^{\circ}$  was employed to take the diffraction pattern of the film. From X-Ray Diffraction pattern, the grain size was calculated using Scherrer's formula

$$D(nm) = \frac{0.9 \cdot \lambda}{B \cdot \cos\theta_{\rm p}}$$

where,  $\lambda$ =0.154 nm is the wavelength of X-Rays corresponding to Cu Ka edge, B the full width half maximum at 20 and  $\theta_B = \theta$ .

Additionally, the presence of thermochromic  $VO_2$  phase was confirmed by Temperature – dependant Micro – Raman spectroscopy using a T64000 J-Y system. The surface morphology was investigated by FE-SEM (Jeol 7000), operating at 15 keV.

Thermochromic properties of VO<sub>2</sub> films were studied by recording the transmittance at elevated temperatures, using a Perkin Elmer Lambda 950 UV/VIS/NIR spectrophotometer at  $\lambda = 250 - 2500$  nm with a homemade heating stage. A thermocouple was in contact with film surface to measure the temperature and a temperature controller was used to control heating by a step of 1.5 °C / min. Transmittance spectra were taken at room temperature (25 °C) and at 90 °C, which is well below and well above T<sub>c</sub> of the material. By using these spectra, both variation of IR transmittance at  $\lambda = 2000$  nm ( $\Delta$ Tr<sub>IR</sub>) and visible transmittance (Tr<sub>vis</sub>) were calculated. The former defined as the difference in transmittance at  $\lambda$ = 2000 nm between 25 °C and 90 °C, while the latter is defined as the transmittance of the film at  $\lambda = 600$ nm. Moreover, integrated luminous transmittance ( $Tr_{lum}$ , 350–750 nm and solar transmittance ( $Tr_{sol}$ , 250–2500 nm) were obtained from the measured spectra, using the equation.

$$Tr_{i} = \frac{\int \varphi_{i}(\lambda) \cdot Tr(\lambda)}{\int \varphi_{i}(\lambda)},$$

where,  $Tr(\lambda)$  denotes the transmittance at wavelength  $\lambda$ , i denotes luminous (lum) or solar (sol) for calculations,  $\varphi_{lum}$  is the standard luminous efficiency function for photopic vision [**26**], and  $\varphi_{sol}$  is the solar irradiance spectrum for an air mass of 1.5 (corresponding to the sun standing 37 ° above the horizon) [**27**]. From this, the solar modulation is defined as the difference  $\Delta Tr_{sol} = Tr_{sol} (25^{\circ}C) - Tr_{sol} (90^{\circ}C)$ , of integrated solar transmittance between 25 °C and 90 °C.

Finally, by taking the transmittance hysteresis loop at  $\lambda = 2000$  nm, between 25 °C and 90 °C, both critical transition temperature (T<sub>c</sub>) and width of transmittance hysteresis loop ( $\Delta T$ ) were extracted. In particular, by plotting the derivative of transmittance (dTr/dT) versus temperature for heating and cooling procedure and by fitting a Gaussian curve, the transition temperatures (T<sub>1</sub>) and (T<sub>2</sub>) were calculated, respectively. Thus, the critical temperature (T<sub>C</sub>) is defined as T<sub>c</sub> = (T<sub>1</sub>+T<sub>2</sub>)/2 and the width of transmittance hysteresis loop defined as  $\Delta T = T_1-T_2$ .



Fig. 1. (a) XRD pattern and (b) T – dependent Raman measurements of a 80 nm  $VO_2$  thermochromic film deposited on flexible Corning® Willow® glass substrate.

## **Results and discussion**

All VO<sub>2</sub> films were polycrystalline as revealed by XRD analysis presented in (**Fig.1a**). According to this, a preferable growth orientation of the (011) plane, which is characteristic for VO<sub>2</sub> and corresponds to  $2\theta = 27.77^{\circ}$ (JCPDS card No. 44025) was obtained. Using Scherrer's formula, grain size was calculated and found to be D = 10.2 nm which is one of the lower values referred to the literature [**28**, **29**]. The presence of thermochromic VO<sub>2</sub> was verified by T-dependent micro Raman spectroscopy (**Fig.1b**). The spectrum was recorded gradually elevating the temperature from RT to 70 °C and back to RT [**21**]. Peaks of 190 and 223 cm<sup>-1</sup> correspond to V-V vibration modes, while that of 620 cm<sup>-1</sup> corresponds to V-O vibration modes and are the monoclinic VO<sub>2</sub> signatures [**25**].

Furthermore, the decrease of 190 cm<sup>-1</sup> peak intensity upon heating is an indication of the phase transition of VO<sub>2</sub> from low temperature, monoclinic, to high temperature, tetragonal rutile, for which the enhanced symmetry leads to the disappearance of the 223 and 620 cm<sup>-1</sup> Raman phonon modes [**30**] at around 50 °C. However, a more accurate account of the film's T<sub>C</sub> evaluation based on the transmittance hysteresis loop is presented above.

The surface morphology of the films was examined by SEM, an image of which is presented in (**Fig.2**). It can be seen that films' surface is very smooth with small grains, in agreement with XRD measurements.



Fig. 2. SEM image of a 80 nm  $VO_2$  thermochromic film deposited on flexible Corning® Willow® glass substrate.

In (Fig. 3a), the transmittance at T = 25 °C and T = 90 °C, of the 80 nm VO<sub>2</sub> film deposited on flexible glass is presented. It can be clearly seen that the film exhibits a thermochromic behavior having an IR transmittance variation at  $\lambda = 2000$  nm of  $\Delta Tr_{IR} = 36\%$  (from 48% at RT to 12% at 90 °C) resulting to a modulation of the integrated solar transmission  $\Delta Tr_{sol} = 5\%$ , which is a typical value for undoped VO<sub>2</sub> films, deposited on various rigid substrates with or without a buffer layer, as it can be seen in **Table 1**. The corresponding transmittance hysteresis loop at  $\lambda = 2000$  nm is presented in (**Fig. 3b**), from which the critical

transition temperatures  $T_1$  and  $T_2$  during heating and cooling, respectively, were calculated and found to be 56.8 °C and 44.7 °C. The former is in good agreement with the value extracted by the above mentioned Tdependent micro Raman spectroscopy (**Fig. 1b**) where the peaks corresponding to VO<sub>2</sub> disappeared at 56 °C, during heating from RT to 70 °C. The width of the transmittance hysteresis loop  $\Delta T$  was 12.1 °C, while the critical transition temperature  $T_c$  was 50.7 °C, which is 18 °C below that is calculated for a single crystal of VO<sub>2</sub>.



Fig.3. (a) Transmittance spectra at different temperatures and (b) transmittance hysteresis loop at  $\lambda = 2000$  nm, of a 80 nm VO<sub>2</sub> thermochromic film deposited on flexible Corning® Willow® glass substrate.

Taking into account that the films were not doped and were grown at low deposition temperature without further annealing, such a low T<sub>c</sub> can be attributed to the inherent small crystallite size and the resulting increased mechanical internal stresses. This is in agreement with Suh et al. [31] who reported that T<sub>c</sub> is size-dependent. Moreover, these values are close enough to those calculated by Melnik et al. [17] for VO<sub>2</sub> films deposited on rigid glass at 200 °C, however, followed by postdeposition annealing at 300 °C. Evidence, that the small grain size leads to transmittance hysteresis width slightly over 10 °C, has also been reported recently by Zhang et al. [32]. It is known [29] that small crystallite size implies high density of grain boundaries resulting in more defects that can cause a distortion by lowering the distance between V-V pairs and as a consequence lower energy is needed in order to change the crystallographic phase from monoclinic to tetragonal rutile structure thus decreasing the critical transition temperature. On the other hand, the lack of sharp transmittance hysteresis loop can be also attributed to defects that make the transition difficult, and slow the propagation of transition during heating and cooling procedure. Finally, a slight increase of 4% at  $\lambda = 600$  nm for the visible transmittance was observed (rising up to 40%); however the integrated luminous transmittance was slightly decreased from 34.14% at 25 °C to 33.71% at 90 °C. This is in agreement with results from Xu *et al.* [**33**], reporting that the integrated luminous transmittance (Tr<sub>lum</sub>) is thickness dependent and that Tr<sub>lum</sub> for films with thickness over 50 nm decreases after heating.

**Table 1.** Thermochromic and optical properties of undoped VO<sub>2</sub> films deposited on various glass substrates referred in literature. T<sub>c</sub> is the critical transition temperature,  $\Delta T$  is the width of transmittance hysteresis loop, Tr<sub>lum</sub>, is the integrated luminous transmittance at 25 °C and  $\Delta Tr_{sol}$  is the solar modulation.

deposition technique	substrate	Т <sub>С</sub> (°С)	ΔT (°C)	ΔTr <sub>sol</sub> (%)	Tr <sub>lum</sub> (%)	Ref.
rf sputtering	Flexible Glass	50.7	12.1	5	34	this work
rf sputtering	SiO <sub>2</sub> /Glass	57	15.8	6.3	30.1	[22]
rf sputtering	SnO <sub>2</sub> /Glass	55.7	8.2	5.2	36.2	[20]
rf + dc sputtering	Glass	59	3	12.8	24.7	[34]
Sol-Gel	Glass	62.5	5	4.54	10.06	[12]
hydrothermal	Plastic (PET)	55.9	8.7	13.6	29.2	[24]

To summarize, this was a first successful attempt to deposit undoped VO<sub>2</sub> on flexible glass, at low substrate temperature of 300 °C, using rf sputtering technique. Both thermochromic and luminous characteristics of the films were comparable with those deposited on rigid or plastic substrates with post annealing, as it can be seen in **Table 1**. In specific, VO<sub>2</sub> films on flexible substrate have lower  $T_C$  and higher luminous transmittance than films deposited on rigid or other substrates. However, width of transmittance hysteresis loop is over 10 °C, while solar modulation is lower than the others. These variations between films deposited in different substrates can be attributed to different thickness, grain size or deposition temperature.

# Conclusion

Thermochromic VO<sub>2</sub> films were deposited directly on flexible glass substrate for first time, at a low substrate temperature of 300 °C, with no post-deposition annealing. Films found to be polycrystalline, with small crystallite size of 10 nm. The thermochromic behavior of the films was verified by both T-dependent micro Raman spectroscopy and transmittance measurements upon heating. They exhibited a low critical transition temperature of  $T_c = 51$  °C, while the width of transmittance hysteresis loop was  $\Delta T = 12$  °C. Both values were attributed to the small crystallite size, due to the low deposition temperature. Additionally, IR transmittance variation ( $\Delta T_{TIR}$ ) at  $\lambda = 2000$  nm was decreased by 36% upon heating, leading to a modulation of integrated solar transmission  $\Delta Tr_{sol} = 5\%$ , as a consequence of film thickness and grain size too. Finally, integrated luminous transmittance  $Tr_{lum}$  was almost unaffected maintaining a value of around 34%, thus demonstrating the potential of using these thermochromic films of low deposition temperature on flexible substrates for smart windows applications. A comparison with previous works on thermochromic VO<sub>2</sub> deposited on rigid glass or plastic substrates was done, resulting that flexible substrate is an appropriate candidate in order to deposit thermochromic VO<sub>2</sub> films.

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#### Author's contributions

Conceived the plan: GK, EA, EG, VB, YR, DT; Performed the expeirments: EG, GM, IK, MP; Data analysis: EG; Wrote the paper: EG. Authors have no competing financial interests.

#### References

- Granqvist, CG.; CRC Crit. Rev. Solid State Sci., 1990, 16, 291. DOI: 10.1080/10408439008242184
- Kiri, P.; Hyett. G.; Binions, R.; Adv. Mater. Lett., 2010, 1, 86. DOI: <u>10.5185/amlett.2010.8147</u>
- Kamalisarvestani, M.; Saidur, R.; Mekhilef, S.; Javadi, FS.; *Renewable and Sustainable Energy Rev.*, 2013, 26, 353. DOI: 10.1016/j.rser.2013.05.038
- 4. Verleur, HW.; Barker, AS.; Berglund, JR. and CN.; *Phys. Rev.*, **1968**, *172*, 788.
- **DOI:** <u>http://dx.doi.org/10.1103/PhysRev.172.788</u> 5. Goodenough, JB.; *J. of Solid State Chem.*, **1971**, *3*, 490.
- DOI: 10.1016/0022-4596(71)90091-0
- Li, S-Y.; Niklasson, GA.; Granqvist, CG.; *Thin Solid Films*, 2012, 520, 3823.
- DOI: <u>10.1016/j.tsf.2011.10.053</u>
  7. Wentzcovitch, R.; Schulz, W.; Allen, P.; *Phys. Rev. Let.*, **1994**, *72*, 3389.
- DOI: 0031-9007/94/72(21)/3389(4)
  Okazaki, K.; Sugai, S.; Muraoka, Y.; Hiroi, Z.; *Phys. Rev. B*, 2006, 73, 165116.
  - **DOI:** 10.1103/PhysRevB.73.165116
- 9. Kivaisi, R.; Samiji, M.; Sol. Energy Mater. Sol. Cells, 1999, 57, 141.

**DOI:** <u>10.1016/S0927-0248(98)00166-4</u>

- Babulanam, SM.; Eriksson, TS.; Niklasson, GA.; Granqvist, CG.; Sol. Energy Mater., 1987, 16, 347.
   DOI: <u>10.1016/0165-1633(87)90029-3</u>
- Vernardou, D.; Louloudakis, D.; Spanakis; E.; Katsarakis, N.; Koudoumas, E.; Sol. Energy Mater. Sol. Cells, 2014, 128, 36. DOI: <u>http://dx.doi.org/10.1016/j.solmat.2014.04.033</u>
- Wang, N.; Magdassi; S.; Mandler, D.; Long, Y.; *Thin Solid Films*, 2013, 534, 594.
   DOI: <u>http://dx.doi.org/10.1016/j.tsf.2013.01.074</u>
- 13. Jian, J.; Chen, A.; Zhang, W.; Wang, H.; J. Appl. Phys., 2013, 114, 244301.

**DOI:** <u>0021-8979/2013/114(24)/244301/6</u>

14. Jin, P.; J. Vac. Sci. Technol., A, 1997, 15, 1113.

DOI: 10.1116/1.580439

- 15. Ruzmetov, D.; Zawilski, KT.; Narayanamurti, V.; Ramanathan ,S.; J. Appl. Phys., 2007, 102, 113715. DOI: 10.1063/1.2817818
- 16. Dinh, NN.; Thao, TT.; Thuc, VN.; Thuy, NTT.; VNU Journal of Science, Mathematics-Physics, 2010, 26, 201.
- 17. Melnik, V.; Khatsevych, I.; Kladko, V.; Kuchuk, A.; Nikirin, V.; Romanyuk, B.; Mater. Lett., 2012, 68, 215. DOI: 10.1016/j.matlet.2011.10.075
- 18. Luo, Y. Y.; Zhu, L. Q.; Zhang, Y. X.; Pan, S. S.; Xu, S. C.; Liu, M.; Li, G. H.; J. Appl. Phys., 2013, 113, 183520. DOI: http://dx.doi.org/10.1063/1.4803840
- 19. Kim, H.; Charipar, N.; Osofsky, M.; Qadri, S. B.; Piqué, A.; Appl. Phys. Lett., 2014, 104, 081913.
- DOI: http://dx.doi.org/10.1063/1.4866806 20. Gagaoudakis, E.; Kortidis, I.; Michail, G.; Tsagaraki, K.; Binas, V.; Kiriakidis, G.; Aperathitis, E.; Thin Solid Films, 2015, 601, 99. DOI: 10.1016/j.tsf.2015.11.007
- 21. Panagopoulou, M.; Gagaoudakis, E.; Aperathitis, E.; Michail, I.; Kiriakidis, G.; Tsoukalas, D.; Raptis, YS.; Thin Solid Films, 2015, 594, 310. DOI: 10.1016/j.tsf.2015.06.010
- 22. Zhu, B.; Tao, H.; Zhao, X.; Inf. Phys. Tech., 2016, 75, 22. DOI: http://dx.doi.org/10.1016/j.infrared.2016.01.004
- 23. Gao, Y.; Luo, H.; Zhang, Z.; Kang, L.; Chen, Z.; Du, J.; Kanehira, M.; Cao, C.; Nano Energy, 2012, 1, 221. DOI: 10.1016/j.nanoen.2011.12.002
- 24. Gao, Y.; Wang, S.; Luo, H.; Dai, L.; Cao, C.; Liu, Y.; Chen, Z.; Kanehira, M.; Energy Env. Sci., 2012, 5, 6104 DOI: 10.1039/c2ee02803d
- 25. Zhang, D-P.; Zhu, M-D.; Liu, Y.; Yang, K.; Liang, G-X.; Zheng, Z-H.; Cai, X-M.; Fan, P.; J. Alloys Compd., 2016, 659, 198. DOI: 10.1016/j.jallcom.2015.11.047
- 26. Wyszecki, G.; Stiles, W.S.; Color Science: Concepts and Methods, Quantitative Data and Formulae; 2nd edition Wiley: New York, NY, USA, 2000. ISBN: 978-0471399186
- 27. ASTM G173-03 Standard Tables of Reference Solar Spectral Irradiances: Direct Normal and Hemispherical on a 37° Tilted Surface; Annual Book of ASTM Standards, American Society for Testing and Materials, Philadelphia, PA, USA, 2003 Vol. 14.04, http://rredc.nrel.gov/solar/spectra/am1.5
- 28. Brassard D, Fourmaux S, Jean-Jacques M, Kieffer JC, El Khakani MA, Appl. Phys. Lett., 2005, 87, 051910. DOI: 10.1063/1.2001139
- 29. Miller, MJ.; Wang, J.; J. Appl. Phys., 2015, 117, 034307. DOI: <u>10.1063/1.4906122</u>
- 30. Pan, M.; Liu, J.; Zhong, H.; Wang, S.; Li, Z.; Chen, X.; Lu, W.; J. Cryst. Growth, 2004, 268, 178. DOI: 10.1016/j.jcrysgro.2004.05.005
- 31. Suh, JY.; Lopez, R.; Feldman, LC.; Haglund, RF Jr.; J. Appl. Phys., 2004, 96, 1209.
- DOI: 10.1063/1.1762995 32. Zhang, H.; Wu, Z.; Wu, X.; Yang, W.; Jiang, Y.; Vacuum, 2014, 104.47.

DOI: 10.1016/j.vacuum.2014.01.003

- 33. Xu, G.; Jin, P.; Tazawa, M.; Yoshimura, K.; Appl. Surf. Sci., 2005, 244, 449.
  - DOI: 10.1016/j.apsusc.2004.09.157
- 34. Choi, Y.; Jung, Y.; Kim, H.; Thin Solid Films 2016, 615, 437. DOI: <u>10.1016/j.tsf.2016.07.051</u>





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