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Comparison studies of SiO₂ and HCI-SiO₂ filler on the film formation and ionic conductivity of <u>PMMA/ENR 50 electrolytes</u>

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ABSTRACT

In this study, polymer electrolytes derived from polymethyl methacrylate/50 % epoxidized natural rubber (PE) were successfully prepared. The effects of silicon dioxide (SiO₂) (15 nm) and acid modified SiO₂ (HCI-SiO₂) (15 nm) on the film formation and ionic conductivity of polymer electrolytes were investigated. In samples preparation, SiO₂ was used as received. Meanwhile, HCI-SiO₂ was prepared by reacting SiO₂ with 7 M of hydrochloric acid (HCl) solution. PE, PE filled with SiO₂ (PE-SiO₂) and PE filled with HCI-SiO₂ (PE-HCI-SiO₂) electrolytes were prepared by solvent casting method with tetrahydrofuran (THF) as solvent and lithium tetrafluoroborate (LiBF₄) as doping salt. The smoothness surface, opacity and agglomeration of PE-HCI-SiO₂ film were found in between PE and PE-SiO₂. CHNS analysis shows that the percentage of hydrogen in HCI-SiO₂ and SiO₂ are 0.88 and 2.07 %, respectively. Fourier transform infra-red (FT-IR) analysis confirm that HCI-SiO₂ has low number of silanol group (Si-OH) but high number of siloxane group (Si-O-Si) compared to SiO₂. Field Emission Scanning Electron Microscopy (FESEM) analysis shows that HCI-SiO₂ has smaller size of agglomeration and porosity compared to SiO₂. Meanwhile, electrochemical impedance spectroscopy (EIS) analysis shows that the ionic conductivity of PE-HCI-SiO₂ is higher than PE but slightly lower than PE-SiO₂. Copyright © 2016 VBRI Press.

Keywords: Silicon dioxide; silanol group; polymethyl methacrylate; epoxidized natural rubber; polymer electrolyte.

Introduction

Studies on development of polymer electrolytes have been given big attention by researchers since it was firstly discovered in 1970s by Wright [1]. Since few decades ago, the early period of development of polymer electrolytes was only focused on improvement of ionic conductivity. Lately, focus of development of polymer electrolytes became broader to obtain high energy density, good charge/discharge cyclic stability as well as high safety [2]. These align with the current requirement on designing Li ion batteries include excellence electrochemical performance and down-scale [3].

Polymer electrolytes consist of polymers implanted with inorganic salt in polymer matrix. Polymer electrolytes pleased its properties include free standing film, no leakage, safer, thermally and electrochemically stable [4].

Unfortunately, lower ionic conductivity at lower temperature became the major drawback to prepare a good ionic conductivity. Normally, the cell must be heated up to 100 °C in order to optimize it's performance [5].

Addition of plasticizers was proposed to increase ionic conductivity of polymer electrolytes especially at lower temperature [6]. These plasticized polymer electrolytes known as gelled polymer electrolytes (GPEs). Frequently, plasticizers used were propylene carbonate (PC) [7], dimethyl carbonates (DMC) [8], poly(ethylene glycol) (PEG) [9] and ionic liquids [10]. Unfortunately, GPEs may have disadvantages such as non-free standing, leakage of harmful and high flammable chemical and corrosion to the body of cell [11 - 13].

Addition of inert ceramic fillers were suitable alternative to increase ionic conductivity [14, 15] by creating porosity for moving ions in the polymer electrolytes matrix [16]. This type of polymer electrolytes was classified as solid polymer electrolytes (SPEs). Besides, SPEs were physically stable, free standing and do not require external supporting structure [17]. Unfortunately, most of SPEs were suffer for filler agglomeration [18 - 20]. Usually, the surface of film becomes rough, brittle and opaque. Therefore, SPEs has less contact with electrodes and limits its performance [21].

In this study, for the first time, we presented the comparison effects of SiO_2 (15 nm) and HCl-SiO₂ (15 nm) filler on polymer electrolytes derived from PMMA and

ENR 50. This work was focus on the formation, appearance of free standing and ionic conductivity of polymer electrolytes. The polymer film properties were linked to the nature and functional group on the surface of SiO_2 and HCl-SiO₂.

Experimental

Materials

All materials were used without prior purification. SiO_2 (15 nm), $LiBF_4$ and PMMA (120 000) were purchased from Aldrich Chemistry, USA. ENR 50 kindly supplied from Guthree Polymer. THF was supplied from J.T.Baker, USA and HCl was purchased from Merck, Germany.

Material synthesis

SiO₂ was reacted with 7 M of HCl solution under stirring condition at room temperature for an hour. HCl-SiO₂ slurry was then dried in an oven at temperature of 120 °C for 24 hours without calcination to remove water and excess HCl from the sample. PMMA, ENR 50, SiO₂ and HCl-SiO₂ were dissolved separately in THF (2, 1, 2 and 2% (w/v)) under continuous stirring using magnetic stirrers at room temperature. ENR 50-THF, and fillers-THF solutions were blended with PMMA-THF solution at composition of 10 and 5% (w/w). The mixtures were continuous stirred for 24 hours before LiBF₄ (40% (w/w)) was added. The mixtures were casted in Teflon petri disc under nitrogen flow for 12 hours. Polymer films was then peeled off and dried in an oven at 50 °C.

Characterizations

Digital picture of samples film was captured using DSLR camera model Nikon D7200. The camera was equipped with Tamron 18-270 mm F/3.5-6.3 Di II VC PZD lens. Percentage of hydrogen in SiO₂ and HCl-SiO₂ were determined using Perkin Elmer Series 2 CHNS/O Analyzer 2400. In Tin samples pans, about 2 to 4 mg of samples were applied in combustion tube at temperature of 975 °C. Perkin Elmer Spectrum One FTIR Spectrophotometer was used for FTIR study. The spectra were recorded at room temperature. The samples were scanned at 400-4000 cm⁻¹ with resolution of 16 cm⁻¹ and 4 times scan rate. The FESEM micrographs of SiO₂ and HCl-SiO₂ were recorded using FESEM model Carl Zeiss Supra 40 VP. The samples images were taken under 1000 magnification without prior coated with gold. The impedance of thin film samples was measured using a Hioki 3532-01 LCR High Tester. The instrument was interfaced to a computer and set to measure the impedance in a frequency range of 100 Hz - 1 MHz from 298 up to 353 K. From the Cole-Cole plot obtained, the bulk resistance can be determined hence, the electrical conductivity of the sample can then be calculated using the following equation;

$$\sigma = \frac{t}{R_{b}} \tag{1}$$

where, σ is the ionic conductivity, *t* is the thickness of the sample, A is the contact area between sample and electrode and R_b is bulk resistance.



Fig. 1. Free standing film of un-dope (a) PE and (b) PE-SiO $_2$ and (c) PE-HCl-SiO $_2.$

Results and discussion

The formation of PE, PE-SiO₂ and PE-HCl-SiO₂ films show in **Fig. 1**. The figure clearly shows that PE obtained was free standing film, transparence and mechanically flexible. Unfortunately, PE-SiO₂ film was found opaque, agglomerates and brittle when SiO₂ was added. The opacity of the films was due to the naturally white powder of SiO₂ [**22**]. Meanwhile, the agglomeration phenomenon is normally occurred in polymer filled nano size filler. The filler particles distributed in polymer matrix experience attraction forces between themselves and accumulated [**23**]. Interestingly, the smoothness surface, opacity and agglomeration of PE-HCl-SiO₂ film are in between PE and PE-SiO₂ indicated that HCl-SiO₂ has the ability to hinder formation of attraction force between filler particles.



Fig. 2. Formation of hydrogen bond between silicon dioxide and moisture [20].

Formation of SiO_2 agglomeration as seen as white spots on the surface of the films may due to tendency of oxygen atom of Si-OH on the surface of SiO₂ to form hydrogen bond with moisture in the polymer matrix (**Fig. 2**) [**24**, **25**]. In other words, Si-OH may act like a glue to adhere SiO₂ particles (**Fig. 2**). This due to fact that the hydrogen bond between Si-OH with moisture is stronger than hydrogen bond among water molecules itself [**21**, **26**]. The agglomeration of HCl-SiO₂ in PE-HCl-SiO₂ is expected was improve when the number of Si-OH on the surface of HCl-SiO₂ was reduced by the acid modification. The reduction number of Si-OH on the surface of HCl-SiO₂ after the acid modification was observed by comparing the percentage of hydrogen atom in HCl-SiO₂ and SiO₂ using CHNS analysis as shows in **Fig. 3**. It was found that the percentage of hydrogen in HCl-SiO₂ decreases compared to SiO₂. This proved that the acid modification successfully reduces number of Si-OH on the surface of HCl-SiO₂.



Fig. 3. CHNS analysis corresponding to SiO2 and HCl-SiO2.

FT-IR analysis was used to further confirm the effect acid modification of SiO₂ on the number of Si-OH and Si-O-Si on the surface of SiO₂ and HCl-SiO₂. The acid modification was proven by comparing intensity of peaks related to Si-OH and Si-O-Si include Si-O-Si asymmetric straching at 1104 cm⁻¹, Si-OH straching at 970 cm⁻¹, Si-O-Si bending at 817 cm⁻¹ and Si-O-Si rocking at 474 cm⁻¹ [**27**] (**Fig. 4**). From the result it was found that SiO₂ is higher in Si-OH but lower in Si-O-Si group compared HCl-SiO₂. Contrary, after the acid modification of SiO₂ it was observed that HCl-SiO₂ is lower in Si-OH but higher in Si-O-Si group compared to SiO₂. This observation successfully confirmed that the number of Si-OH on the surface of SiO₂ was reduced by the acid modification.



Fig. 4. FT-IR spectra corresponding to (a) SiO₂ and (b) HCl-SiO₂.

 SiO_2 and $HCl-SiO_2$ were examined under FESEM in order to investigate the effect of acid modification on the size of aggregation of the fillers (Fig. 5). In overall,

aggregation of HCl-SiO2 in **Fig. 5** (b) was observed smaller compared to SiO2 in **Fig. 5** (a).



Fig. 5. FESEM micrograph corresponding to (a) SiO₂ and (b) HCl-SiO₂.

In addition, the figure also shows that the intensity of porosity of HCl-SiO₂ is smaller in size compared to SiO₂. As proposed by Shabanian et al. (2012) [28], the neighboring SiO₂ particles may attract each other due to the attraction forces between them. The attraction force attributed from condensed moisture from the surrounding gas on the surface of SiO_2 in between the two SiO_2 particles. In other words, the moisture may act as a bridge to connecting the two SiO_2 particles [28]. Relatively, the attraction force between the two SiO₂ particles may strongly influence by the number of side available namely as Si-OH on the surface of SiO₂ for the attraction force. The more Si-OH on the surface of SiO_2 , the more attraction force take place in between the two SiO₂ particles. Therefore, it can be concluding that, the modification successfully reduces the number of Si-OH on the surface of SiO₂, minimize the attraction force between the two SiO₂ particles and reduce the SiO₂ agglomeration.

Reaction mechanism during the acid modification of SiO_2 is proposed in **Fig. 6**. The acid acts as a catalyst to reduce the number of Si-OH and increase the number of Si-O-Si group on the surface of HCl-SiO₂, simultaneously. The mechanism was suggested involved in 3 steps and catalyzed by the acid. In the reaction, the acid may protonates the hydroxyl atom of Si-OH. The protonated hydroxyl group is a good leaving group therefore the $Si-OH_2^+$ bond may dissociated to release water molecule [28] in the second step.



Fig. 6. Suggested mechanism of HCl-SiO₂ modification routes [20].

The lone pair electrons of oxygen atom at adjacent Si-OH may strongly attract to O-Si⁺ ion. Therefore, a new bond formed between Si atoms with oxygen atom of adjacent Si-O and releases the acid back. The suggested reaction mechanism successfully demonstrated that the number of Si-OH reduced meanwhile the number of Si-OS increased due to the acid modification [29]. As result, the tendency formation of hydrogen bond between oxygen atoms of Si-OH on the surface of SiO₂ with hydrogen atom of trapped water molecule may successfully depress. The proposed mechanism may use to support CHNS, FT-IR and FESEM analysis result discussed previously in which after the acid modification HCl-SiO₂ compose low in Si-OH.

Room temperature ionic conductivity corresponding to PE, PE-SiO₂ and PE-HCl-SiO₂ electrolytes are shown in **Fig. 7**. The figure shows that ionic conductivity of PE-SiO₂ electrolytes was higher than PE electrolyte. According to Ravi *et al.* (2014), ionic conductivity of polymer electrolytes is depend on concentration of dopant salt and it's mobility [**18**]. Usually, polymer electrolytes with high porosity provide ion mobility and exhibit high ionic conductivity but may decrease its mechanical properties [**30**].



Fig. 7. Ionic conductivity of PE-SiO₂ and PE-HCl-SiO₂.

In addition, the increase in ionic conductivity of $PE-SiO_2$ may attribute to increase the porosity in polymer matrix created by SiO_2 particles [**31**, **32**]. The porosity may

provide easy pathway for Li⁺ ion movement hence increases the ionic conductivity of PE-SiO₂. Contrary, the ionic conductivity of PE-HCI-SiO₂ electrolyte was found decreases compared to PE-SiO₂ electrolyte but still higher than PE electrolyte. This may attributed to the ionic conduction in PE-HCI-SiO₂ are less due to HCI-SiO₂ exhibit smaller porosity compare to SiO₂ as shown in **Fig. 5**.

Conclusion

As conclusion, PE, PE-SiO₂ and PE-HCl-SiO₂ were successfully prepared by solvent casting method. Addition of SiO₂ improved the ionic conductivity of PE-SiO₂ compared to PE. Unfortunately, PE-SiO₂ suffers for SiO₂ roughness surface, opacity, agglomeration and brittleness. Interestingly, addition of HCl-SiO₂ in PE-HCl-SiO₂ had improved the weaknesses found in PE-SiO₂. Nevertheless, PE-HCl-SiO₂ shows slightly lower in ionic conductivity but still higher than PE.

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