

# Effect on Structural, Electrical and Temperature Sensing behavior of Neodymium Doped Bismuth Ferrite

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Synthesis of polycrystalline samples of  $Bi_{1-x}Nd_xFeO_3$  [x = 0.5, 0.6, 0.7 and 0.8] were demonstrated following solid-state reaction method at high temperature. The structural properties of the sample were confirmed through the X-ray diffraction technique. The dielectric study of the compounds was performed at different frequencies in the range of 100 Hz – 10<sup>6</sup> Hz for various temperatures. The non-Debye type of relaxation process confirmed from impedance analysis. The materials showed a negative temperature coefficient of resistance (NTCR) behavior at various temperatures and frequencies. AC conductivity of the materials with frequency at different temperatures satisfied the universal power law of Johnscher. Thermistor constant ( $\beta$ ), sensitivity factor ( $\alpha$ ), and stability factor for all the samples were calculated and confirmed the characteristics of NTC thermistor.

# Introduction

Multiferroic material means the simultaneous existence of two or more Ferroic characteristics namely ferroelectric, ferroelastic or ferromagnetic in the same material [1]. The attention to multiferroic materials is now exponentially increasing due to their unique properties and applications which are discussed in a large no of journals [2,3]. One of the popular multiferroic materials is Bismuth ferrite (BiFeO<sub>3</sub>) where both the ferroelectric and ferromagnetic properties exit simultaneously because of the existence of 6s electron lone pair in Bismuth and the partially filled dorbital of Fe respectively [4]. Due to its tremendous physical properties. It can be employed in a various field specifically in multifunctional electronic devices including spintronics, photovoltaic cell, lithium-ion batteries [5], transistors, electromagnetic interference filters, [6] data storage media, Spin valve devices, non-volatile memory [7], multiple state memories, sensors, transducers, actuators [8], etc. Nowadays, it is a formidable challenge for the researcher and scientists to synthesize and characterize the Bismuth ferrite [9] because of its size effects on the physical properties. There are several synthesis processes to synthesize Bismuth ferrite including the Sonochemical

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method, Chemical method, sol-gel synthesis, solid-state reaction technique, etc. [5,10-12]. But Solid-state reaction technique is cost-effective. Bismuth ferrite has a rhombohedral symmetry and an R3c space group with high Curie temperature (T<sub>C</sub>) at 1083 K and Neel temperature (T<sub>N</sub>) at 673 K [13-15]. There are significant limitations to synthesize single-phase Bismuth Ferrite where current leakage or vacancy of oxygen during sintering is a major problem leading to less ferroelectricity, weak magnetization, in bismuth oxide [16]. To overcome the drawback of the single-phase Bismuth Ferrite, research work has been carried out by doping rare earth elements (La, Gd, Nd, Ca, Sm, V, Eu, etc.) [17,18] at the A - site of the Ferroic materials to enhance the characteristics of these materials [19-21].

Neodymium oxide (Nd<sub>2</sub>O<sub>3</sub>) is a commonly used important rare earth material for the manufacture of glass, capacitors, and magnets. It is reported that DC conductivity increases and saturation magnetization decrease with an increase in the concentration of Nd<sub>2</sub>O<sub>3</sub> with Bismuth Ferrite indicating an ionic contribution to conductivity [22,23]. By doping Nd<sub>2</sub>O<sub>3</sub> with Bismuth Ferrite, leakage current can be reduced and results in the enhancement of electrical and magnetic properties [16]. Also, the resonance frequency will increase with the rising concentration of Nd which can be applicable for radio communication devices [24-26]. At higher temperatures, Nd<sub>2</sub>O<sub>3</sub> has excellent electrical properties [27]. After synthesizing BiFeO<sub>3</sub> by the doping of Neodymium through the sol-gel technique, Dawei Wang concluded that the samples transform from the rhombohedral to orthorhombic phase and curie temperature decreases with changing of the Nd concentration in Bismuth Ferrite [28]. Furthermore, with increasing the content of Nd, the magnetic and electric properties of Bismuth ferrite are enhanced, while the decrease of

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dielectric constant and tangent loss are observed [**29,30**]. As the multiferroic BiFeO<sub>3</sub> materials show NTCR behavior [**31**], these materials can be utilized in various industry including food, chemical, medicine, automobile, etc. These materials also can be utilized as a temperature sensor, measuring instruments and flow of liquid, liquid level regulator, current and voltage limiters, etc [**32**]. In view of the above, we have elaborately discussed the structural, microstructural, Impendence, ac conductivity, and Thermistor study (parameters) of Neodymium doped Bismuth ferrite Bi<sub>1-x</sub>Nd<sub>x</sub>FeO<sub>3</sub> [x = 0.5, 0.6, 0.7and 0.8]

# **Experimental details**

 $Bi_{(1-x)}$  Nd<sub>x</sub> FeO<sub>3</sub>[x = 0.5,0.6,0.7,0.8] were prepared with appropriate stoichiometry content of high purity raw materials i.e.,  $Bi_2O_3$  (M/s. CDH, India > 98%),  $Fe_2O_3$  (M/s. CDH, India, > 98.5%), Nd<sub>2</sub>O<sub>3</sub> (M/s. Spectrochem, India >99.9%) through the process of solid-state reaction. First, the oxides were carefully blended in the presence of air atmosphere for 2 h, and then in alcohol for 1h. The blended powders were calcined for 5h at 800 °C. The calcined powder was then cold-pressed into cylindrical pellets of 10 mm diameter and 1-2 mm of thickness at a pressure of  $3.5 \times 10^6$  N/m<sup>2</sup> using a hydraulic press. Then the pellets were sintered at 850 °C for 6 h in an air atmosphere. For the reduction of the brittleness of the pellet, a binding agent Polyvinyl Alcohol (PVA) was introduced. The phase composition and properties of the samples were examined through the XRD process at room temperature using a powder diffractometer (Rigaku, Ultima IV, Japan) by applying the radiation of CuK<sub> $\alpha$ </sub> of wavelength  $\lambda = 1.5405$  Å in a wide range of Bragg's angles  $2\theta$  ( $20^\circ \le 2\theta \le 80^\circ$ ) with a scanning rate of 3° per minute. To demonstrate the electrical properties of the samples, silver air-dying conducting paste is coated on the sintered pellets. After coating, the pellets were heated at 150 °C for 1 and half hours to remove any moisture and then cooled to room temperature. The impedance parameters were calculated through an LCR meter (HIOKI Model 3532) at a frequency of  $10^2 - 10^6$  Hz and temperatures in the range of 25 - 450 °C.

# **Results and discussion**

### Structural properties

From the XRD graph in **Fig. 1**, the orthorhombic phase with space group Pmna of all the samples  $Bi_{(1-x)}$  Nd<sub>x</sub>FeO<sub>3</sub> [x = 0.5,0.6,0.7,0.8] is confirmed at room temperature. For the detailed structural analysis, the XRD data are refined using a "POWD" computer program package [**33**]. The size of crystallite (P) is approximately determined using Scherrer's equation, P =  $K\lambda/\beta_{1/2} \cos\theta_{hkl}$  [**34,35**], (where K represents dimensionless shape factor which is nearly equal to 0.89,  $\lambda$ represents X-Ray wavelength of 1.5405Å and  $\beta_{1/2}$  = peak width of the broadening line of XRD peak at half of the intensity). The refined lattice parameters and the size ranges of the crystalline are listed in **Table 1**. The average particle size of the samples is in the range of 24-26 nm.



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Fig. 1. X-ray diffraction pattern of  $Bi_{(1-x)}\,Nd_xFeO_3[x=\!0.5,\!0.6,\!0.7,\!0.8]$  and Rietveld Refinement.

Table 1. Comparison of lattice parameters (in Å) and crystal sizes of  $Bi_{(1-x)}Nd_xFeO_3$  [x = 0:5, 0.6, 0.7 and 0.8].

	a (Á)	b (Á)	c (Á)	Chi Square	Size of Crystalline (P)
Bi0.5Nd0.5FeO3	5.4280	7.7879	5.5694	5.37	26
Bi <sub>0.4</sub> Nd <sub>0.6</sub> FeO <sub>3</sub>	5.4397	7.7784	5.5790	4.60	24
Bi0.3Nd0.7FeO3	5.4418	7.7632	5.5810	4.57	25
Bi <sub>0.2</sub> Nd <sub>0.8</sub> FeO <sub>3</sub>	5.4459	7.7651	5.5791	4.04	24

#### Microstructural properties

From the SEM micrographs of the sintered pellet at room temperature, densely packed grains are detected over the surface that confirms the polycrystalline nature of the samples. The increase in the concentration of Neodymium does not affect the nature of the grain as shown in **Fig. 2**. The average grain size of the samples cannot be evaluated because of hazy micrographs.



Fig. 2. Scanning Electron Microscope of  $Bi_{(1x)}Nd_xFeO_3[x = 0.5, 0.6, 0.7, 0.8]$ .

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#### Dielectric study

At different frequencies from the characteristics curve between temperature and the dielectric constant ( $\varepsilon_r$ ), there is a gradual increment of the dielectric constant  $\varepsilon_r$  with rising temperature but decreases with an increase in frequency (**Fig. 3(a)**). The peak height at the transition temperature is observed to decrease with increasing frequency [**36**]. The anomaly is expected to be present in the sample at a higher temperature region. The value of tangent loss (tan $\delta$ ) increases with the increment of temperature for all the materials while decreases with rising frequency (**Fig. 3(b**)). It is recognized that the value of tan $\delta$ increases with the rising content of Nd. This is due to the improvement in conductivity.



Fig 3 Variation of dielectric constant ( $\epsilon_r$ ) and dielectric loss (tan  $\delta$ ) of Bi (1-x) Nd x FeO 3 [x=0.5,0.6,0.7,0.8] with temperature at different frequencies

#### Impedance study

The complex impedance spectroscopy is a unique and powerful tool to investigate the transport characteristics including bulk contribution, grain boundary, and effect of the electrode in the composites [**37**]. The data of electrical ac conductivity for the compounds can be evaluated from the four basic formulas [**38**] which are given as follows:



# Complex impedance: $Z^* = Z' - jZ'' = R - \frac{j}{\omega C}$ , Complex admittance: $Y^* = Y' + jY'' = \frac{1}{R} + j\omega C$ , Complex modulus:

 $M^* = \frac{1}{\varepsilon^*} = M' + jM'' = j\omega C_0 Z, \text{ Complex permittivity:}$  $\varepsilon^* = \varepsilon' - j\varepsilon'', \text{ Loss tangent, } \tan \delta = \frac{\varepsilon''}{\varepsilon'} = \frac{M''}{M'} = -\frac{Z'}{Z''} = \frac{Y'}{Y''}.$ 



Fig. 4. Variation of Z' as a function of frequency at different temperatures of  $Bi_{(1-x)}Nd_xFeO_3[x=0.5,0.6,0.7,0.8]$ .

The curve between the real part of impedance (Z') with the frequency at various temperatures in the range of 200-275 °C is shown in **Fig. 4**. It is noticed that the value of the real part of impedance Z' decreases with the rise in temperature indicating the NTCR characteristics. The values of Z' were combined at high frequencies because of the effect of space charge at high temperatures [**39,40**].



Fig. 5 Variation of Z" as a function of frequency at different temperatures of Bi<sub>(1-x)</sub>Nd<sub>x</sub>FeO<sub>3</sub>[x=0.5,0.6,0.7,0.8].

The variation of imaginary parts of impedance (Z') with frequency is shown in **Fig. 5**. The peak position of all the curves deviates towards the region of high frequency that may be observed because of the bulk resistivity or relaxation process. It also may be due to small polaron hopping with minimization of electron–lattice coupling



[41]. The characteristic curve between the real part (Z') and imaginary part Z" of impedance at various temperatures is comprising of an arc centered below the real axis indicating bulk property as shown in **Fig. 6** and follows non-Debye type behavior.



**Fig. 6** Complex impedance spectrum (Z' vs. Z") and experimental and fitted impedance data of  $Bi_{(1-x)}Nd_xFeO_3[x=0.5,0.6,0.7,0.8]$  at selected temperatures with equivalent circuits (inset).

The Constant Phase Element (CPE) value has been obtained from the Nyquist plot fitting by using ZSimpWin Software which is used in equivalent circuit modeling and data fitting of electrochemical impedance spectroscopy data. The equivalent circuit fitted with R(Q(R(Q(RW)))) as shown in **Fig. 6** for the  $Bi_{1-x}Nd_xFeO_3$  [x = 0.5, 0.6, 0.7 and 0.8] to explain the series of interconnected grain and phenomenon of the grain boundary. The different parameters of the equivalent circuit are given in **Table 2**.

#### Conductivity

From the ac conductivity ( $\sigma_{ac}$ ) vs frequency curve at various temperatures, as shown in **Fig. 7**, the plateaus of ac conductivity ( $\sigma_{ac}$ ) are noticed at low frequencies and high

temperatures. These frequency-independent values of  $\sigma_{ac}$  indicate the dc conductivity which can be determined using the universal power-law equation of Johnscher [42],  $\sigma(\omega) = \sigma_{dc} + A\omega^n$ , where the term  $A\omega^n$  corresponds to the ac dependence describing all the phenomenon of dispersion and the value of 'n' differ from material to material that can be obtained from the fitting parameters as mentioned in **Table 3**.

The ac conductivity ( $\sigma_{ac}$ ) of all the samples rises with an increment of temperatures. At high temperatures, the values of ac conductivity combined indicating frequency-independent i.e. dc conductivity [**43-45**].

Table 2. Values of parameters of equivalent circuit of  $Bi_{(1\cdot x)}Nd_xFeO_3$  [x=0.5,0.6,0.7,0.8] at 200 $^{0}C.$ 

Parameters	X = 0.5	X = 0.6	X = 0.7	X = 0.8
<b>R</b> 1	3.147E+001	1.089E-002	8.542E+001	2.398E+002
CPE or Q1	1.118E-009	8.511E-008	4.083E-010	2.659E-006
Frequency power (n)	8.609E-001	1.132E-008	9.157E-001	1.375E-011
<b>R</b> <sub>2</sub>	3.062E+002	1.903E+002	6.158E+004	1.000E-002
CPE or Q2	2.255E-003	6.414E-008	5.894E-008	8.099E-011
<b>R</b> <sub>3</sub>	4.377E+004	1.016E-001	1.648E+004	7.247E+005
Frequency power (n)	1.928E-011	8.000E-001	6.188E-001	9.928E-001
W	1.746E-007	1.000E-020	5.914E+005	7.693E-009
Chi square	1.077e-04	1.836e-02	4.390e-05	4.974e-04



Fig. 7. Variation of ac conductivity ( $\sigma_{ac}$ ) of  $Bi_{(1-x)}Nd_xFeO_3$  [x = 0.5, 0.6, 0.7, 0.8] with frequency at different temperatures.

Table 3. Fitting parameters obtained from the Jonscher power law at different temperatures.

	B	Bi <sub>0.5</sub> Nd <sub>0.5</sub> FeO <sub>3</sub> Bi <sub>0.4</sub> Nd <sub>0.6</sub> Fe			io.4Ndo.6FeO3	O3 Bi <sub>0.3</sub> Nd <sub>0.7</sub> FeO <sub>3</sub>				Bi <sub>0.2</sub> Nd <sub>0.8</sub> FeO <sub>3</sub>		
T <sup>0</sup> C	$\sigma_{dc}$	А	n	$\sigma_{dc}$	А	n	$\sigma_{dc}$	А	n	$\sigma_{dc}$	А	n
200°C	5.00206E-4	8.7241E-8	0.75321	2.09642E-5	2.36381E-9	0.8965	-0.00706	9.06688E-5	0.52222	5.00039E-5	3.52014E-11	1.21169
225°C	7.43398E-4	3.84167E-7	0.64961	4.05697E-5	1.58816E-9	0.93978	1.70261E-4	1.84925E-9	0.98512	1.15061E-4	3.26417E-10	1.04379
250°C	8.20989E-4	4.59889E-5	0.36917	7.36679E-5	3.27942E-8	0.74005	2.2378E-4	5.37037E-9	0.90652	1.55704E-4	7.83372E-9	0.81649
275°C	-0.00534	0.00187	0.2095	0.00163	2.07094E-8	0.85493	2.96146E-4	5.36138E-8	0.74195	1.90713E-4	3.43872E-7	0.56218

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Fig. 8. Variation of ac conductivity  $(\sigma_{ac})$  with  $(10^3/T)$  of  $Bi_{(1-x)}Nd_xFeO_3[x=0.5,0.6,0.7,0.8].$ 



**Fig. 9.** Variation of Resistance R with Temperature T and logarithmic R with Temperature T of  $Bi_{(1:x)}Nd_xFeO_3[x = 0.5, 0.6, 0.7, 0.8]$ .

#### Electrical resistance

The electrical resistance of samples  $Bi_{(1-x)}$  Nd<sub>x</sub> FeO<sub>3</sub>[x=0.5,0.6,0.7 and 0.8] are determined at different temperatures. The exponential variation of resistance of the

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samples with different temperatures ranging from 175°C to 250°C confirms the NTCR behavior [46,47] of samples as shown in Fig. 9(a). It is observed that the value of resistance increases with the increase of Nd concentration in BFO and indicates the transformation of the state of conductive to insulating state [48].

Electrical resistance (R) of NTC thermistors can be explained as a function of temperature which is related by the equation  $R_T = \exp(\frac{1}{T})$ . It can be also simplified as  $ln(R_T) = \frac{1}{T} + C$ , where  $R_T$  represents the resistance of the sample at respective temperature T. The variation between logarithmic resistance and the inverse temperature is linear as shown in **Fig. 9(b)**. This linearity behavior of samples explained the excellent NTCR characteristics that can be useful in the high-temperature thermistor including the temperature sensor. The deviation in graphs for some samples was aroused because of systematic errors of processing parameters, the effect of sintering, and doping contents.

#### Thermistor constant (β)

Thermistor constant  $(\beta)$  can be determined from the relation between electrical resistance and temperature which is

given by 
$$\beta = \frac{ln\left(\frac{R_1}{R_2}\right)}{\frac{1}{T_1} - \frac{1}{T_2}}$$
. It is found that NTCR

thermistors follow the equation of Arrhenius in the temperature range of 175 - 250 °C. The  $\beta$  values of all the compositions at different temperatures are mentioned in **Table 3**.

For NTC thermistor, the thermistor constant ( $\beta$ ) is closely related to electrical resistance (R) and temperature (T) by the relation  $R = exp\left(\frac{\beta}{T}\right)$ . The above equation indicates linearity characteristics between thermistor constant  $\beta$  and the natural logarithm resistance (R) satisfying the Arrhenius equation [32] as noticed in **Fig. 10**. Because of these linear relations, it can be useful for biomedical and medical applications, thermistor-based devices, temperature sensors, etc. [29].



Fig. 10. Variation of logarithmic of resistance with  $\beta(K)$  of  $Bi_{(1-x)}Nd_xFeO_3[x = 0.5, 0.6, 0.7, 0.8]$ .

**Table. 4.** Concentration dependent thermistor constant( $\beta$ ), activation energy( $E_a$ ) and sensitivity index( $\alpha$ ) of BNF ceramic.

		Sintering Temperature 850°C										
Measuring temperature	В	i0.5Nd0.5FeC	)3	]	Bi0.4Nd0.6Fe	03	Bi <sub>0.3</sub> Nd <sub>0.7</sub> FeO <sub>3</sub>		В	Bi <sub>0.2</sub> Nd <sub>0.8</sub> FeO <sub>3</sub>		
(°C)	β	Ea	α	β	Ea	α	β	Ea	α	β	Ea	α
175	8417.895	0.7254	-0.0470	12563.23	1.08261	-0.07021	10326.47	0.8898	-0.05771	9414.911	0.8113	-0.05262
200	6721.996	0.5792	-0.0334	9386.672	0.80888	-0.04677	9887.316	0.8520	-0.04926	7901.24	0.6808	-0.03937
225	4209.274	0.3627	-0.0188	4996.794	0.43059	-0.02233	7013.656	0.6043	-0.03135	6703.36	0.5776	-0.02996
250	3647.04	0.3142	-0.0147	3851.207	0.33187	-0.01553	5536.62	0.47711	-0.02232	6576.421	0.56671	-0.02652

#### Temperature coefficient of resistance

The value of sensitivity index or temperature coefficient of resistance ( $\alpha$ ) can be determined from the value of thermistor constant  $\beta$  using the relation of Steinhart–Hart coefficient  $\beta = \sqrt{\left(\frac{b}{3c}\right)^3 + \frac{\alpha^2}{4}}$  [46] which can be simplified as  $\alpha = \frac{a-T}{c} = \frac{-\beta}{T^2}$ . It is confirmed that  $\alpha$  decreases with rising in thermistor constant  $\beta$  [49] as shown in Table 3. The sensitivity index ( $\alpha$ ) is generally described in percentage per degree centigrade (%/°C) [47].

#### Sensitivity

The sensitivity of an NTC thermistor can be explained by its sensitivity index ( $\alpha$ ) and activation energy (E<sub>a</sub>) as shown in **Table 4** for the different content of dopant Nd in BFO. The sensitivity of the composites decreases up to x=0.6 in Bi<sub>1-x</sub>Nd<sub>x</sub>FeO<sub>3</sub>; but it rises unexpectedly at x = 0.7. These variations of sensitivity may be due to the existence of extrinsic charge carriers in grain.

#### Stability factor

The stability factor can be determined by using simple relation which is equal to the ratio of highest to the lowest value of resistance. As the content of dopant Nd rises in BFO, the stability factor of all the materials decreases. Based on the above explanation, it is confirmed that Nd-doped BFO is a suitable material for NTC thermistors with better performance and stability factors as shown in **Table 5**.

**Table 5.** Stability factor at different concentration of Nd.

Doping concentration of Nd	$R_{max}/R_{min}$	Stability factor $log(R_{max}/R_{min})$
x = 0.5	92.9802	1.9683
x = 0.6	58.72616	1.76883
x = 0.7	43.989308	1.64334
x = 0.8	30.9588	1.490785

# Conclusion

 $Bi_{(1-x)} Nd_x FeO_3[x=0.5,0.6,0.7 and 0.8]$  composites were synthesized using a solid-state reaction technique and the orthorhombic structure with space group Pmna. The average crystallite sizes of the samples are in the range of 24-26 nm. The dielectric constant of all the samples increases with a rise in temperature. The Nyquist plot reveals the contribution of the bulk characteristics in the materials. AC conductivity study with frequency obeys the power law. The linearity relationship between logarithmic resistance and the inverse temperature is observed that confirms the NTC thermistor properties. On the increment of dopant Nd content, its stability factor decreases. All the materials showed NTCR behavior. Due to the above excellent NTCR behavior, it can be applicable in temperature sensors, thermistor-based electrical devices, etc.

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#### **Conflicts of interest**

There are no conflicts to declare.

#### Keywords

Solid-state reaction, XRD, electrical conductivity, thermistor constant, NTCR behavior.

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